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**PHYSICO-CHEMICAL ASPECTS OF TERNARY AND COMPLEX PHASES  
DEVELOPMENT BASED ON THALLIUM CHALCOHALIDES**

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*The results of physical and chemical investigation of some ternary and more complex systems which are of interest in terms of obtaining thallic chalcogenides and new multicomponent phases on their basis have been systematized and analyzed in the review paper. This class of inorganic substances is of scientific and practical interest as promising functional materials with thermoelectric, optical, magnetic, sensor and detector properties. The literature data on crystal structure, thermodynamic and some physico-chemical properties of thallium chalcogenides and intermediate phases on their basis have been investigated.*

**Keywords:** *thallium chalcogenides, functional materials, phase diagram, solid solutions, thermodynamic properties, crystal structure.*

## 1. Introduction

Chalcogenides and chalcogenides of heavy metals, including 5p and 6p elements, are in the centre of attention of world scientists since the middle of the last century due to their interesting chemical bonding characteristics and fascinating physical properties [1-5]. Many of them are applied or considered perspective for application in various fields of electronics and high technologies.

The discovery of research achievements in nanomaterials over the past two-three decades [6,7] and the discovery of graphene [8] and topological insulators (TI) [9-11] have given a powerful impetus to the development of electronics and other high technologies. The discovery of graphene stimulated the search for other graphene-like two-dimensional materials, among which 2D transition metal chalcogenides, due to their unique physical properties, proved extremely promising for use in new generation optoelectronic devices [12-16].

The most important applications of TI materials are new spin-off devices and permanent transistors for quantum computers, based on the quantum spin Hall effect [9-11, 17] and the anomalous quantum Hall effect [18]. These materials are also prospective for application in superior magnetoelectronics, optoelectronics, security systems, and so on

[19-21]. Currently the most widely studied substances like TI are Bi<sub>2</sub>Se<sub>3</sub>, Bi<sub>2</sub>Te<sub>3</sub>, Sb<sub>2</sub>Te<sub>3</sub> compounds and phases based on them [9-11, 22, 23]. However, recent studies have shown that a number of tetradimide-based compounds formed in A<sup>IV</sup>-B<sup>V</sup>-Te systems also possess interesting TI properties [11, 24-27].

Heavy metal chalcogenides play an important role among the new photoelectric, thermoelectric materials and other energy-transducers generated for the use of alternative and renewable energy sources [28-31]. These materials are also used to create highly sensitive sensors and detectors for the implementation of various complex technological processes, as well as, for monitoring environmental safety on Earth in the atmosphere [7, 32].

Chalcogenides of heavy metals, including 5p- and 6p-elements occupy a special place among the above mentioned promising materials for application in many other areas [3, 33-43]. For example, ferroelectric semiconductors of type B<sup>V</sup>XHal (B<sup>V</sup>-Sb, Bi, X-S, Se, Te, Hal, Cl, Br, I) possess precious photoelectric, magnetic, electrooptic, pyroelectric, pyro-optic and other properties and considered to be promising for application as nuclear,  $\gamma$ - and infrared beam

detectors, light modulators, gas sensors, photon crystals, memory elements, etc.[33-35].

It is expected that the Rashba effect, which provides spin manipulation through the electric fields of these materials, will be used in devices proposed as a spin field and spin Hall transistors [36-40]. The most recent materials proposed in this direction are, for example, spin-orbit coupling, ferroelectric semiconductors controlled by electric polarization [41-43].

Over the last decade, the phase equilibrium, thermodynamic properties of  $B^V-X-I$  systems have been studied extensively, primary crystallization areas and fundamental thermodynamic functions of intermediate phases, including  $B^VXI$  compounds, have been determined [44-48].

Numerous studies have shown that ternary chalcogenides and chalcogenides of 6p (Tl, Pb, Bi) and 5p (Sn, Sb) elements have an anomalous low thermal conductivity and high thermoelectric efficiency [29-31].

According to recent studies, thallium chalcogenides and ternary chalcogenide hybrids ( $Tl_6Se_4, Tl_6SI_4$ ) have potential applications, such as radiation detectors, and form a new class of inorganic semiconductor detector materials [49-57]. These compounds and phases on their basis are far more promising than CZT ( $Cd_{0.9}Zn_{0.1}Te$ ) of industrial importance - the most sensitive detector material for functional indications, and are one of the most promising materials for new generation electronic devices. Even though the thallium compounds are toxic, compounds and intermediate phases containing thallium have an important place among such materials that have unusual electronic properties. It has been established that three-dimensional seleno- and thioiodides of thallium have three-dimensional structure (3D) and therefore possess better internal electronic and mechanical properties, demonstrating both

high hole and electron conductivity compared to linear (1D) and layered (2D) compounds [49-52].

$Tl_6S(Se)_4$  monocrystals were studied, their electron structure, dielectric, optical properties and structure defects examined [53-54]. It has been established that these ternary compounds have higher mechanical strength and optimal band gap value for radiation detectors compared to TII and  $Tl_2S(Se)$  binary compounds that form them.

Note that the establishment of scientific basis for the directional search and synthesis of new multicomponent inorganic materials with a combination of physical and physicochemical properties requires a complex analysis of the corresponding systems, their phase diagrams, and the determination of the fundamental thermodynamic functions of the intermediate phases [58, 59]. From this point of view, it is of particular interest to have a complex physical-chemical study of complex systems known to be formed of solid analogues or solid solutions based on known binary and triple compounds with valuable application properties. Therefore, a physicochemical investigation of complex systems composed of structure analogs of known binary and ternary compounds with valuable application properties or solid solutions based on them is of particular interest.

The review article summarized and comparatively analyzed the literature data and experimental results obtained by the authors on the ternary  $Tl-X-Hal$  systems,  $Tl_2Te-TlHal-TlHal'$  and  $Tl_5Te_3-Tl_9BiTe_6-Tl_5Te_2Hal$  concentration ranges of the  $Tl-X-X'-Hal$  or  $Tl-X-Hal-Hal'$  ( $X, X' - S, Se, Te$ ;  $Hal, Hal' - Cl, Br, I$ ) quaternary systems, and on the phase equilibria, thermodynamic, crystallographic and some physicochemical properties of the complex thallium chalcogenide systems.

## 2. Phase equilibria in the $Tl-X-Hal$ ( $X-S, Se, Te, Hal -Cl, Br, I$ ) systems and some physico-chemical properties of intermediate phases

Initial researches on  $Tl-X-Hal$  ( $X-S, Se, Te, Hal -Cl, Br, I$ ) systems began in the 1980-s

with the study of  $Tl_2X-TlHal$  and  $TlX-TlHal$  quasibinary sections.

2.1. Phase equilibria in the Tl – S – Cl (Br,I ) systems

**Tl-S-Cl system.** TlCl-Tl<sub>2</sub>S quasi-binary section of this system is of peritectic type [60, 61] and forms a congruently melted Tl<sub>6</sub>SCl<sub>4</sub> ternary compound at 683 K (Fig. 1, a). In [62], the system was re-visited, formation of the Tl<sub>6</sub>SCl<sub>4</sub> peritectic ternary compound was confirmed and coordinates of the peritectic and eutectic equilibria were found to be essentially different from the results of the [60,61] (Fig.1, b). In contrast to [60,61], existence of the immiscibility area (~13-50 mol% Tl<sub>2</sub>S) of two liquid phases has been

established [62].

The system was studied at the Tl-TlCl-S concentration range by means of DTA, X-ray and SEM analysis, as well as through the measurement of microhardness and EMF. A series of polythermal and isothermal sections, liquidus surface projection of its phase diagram was constructed [62-64].

It found that TlCl-S system is also a quasibinary, forming a phase diagram with a monotectic and degenerated eutectic equilibrium near the sulfur [63].

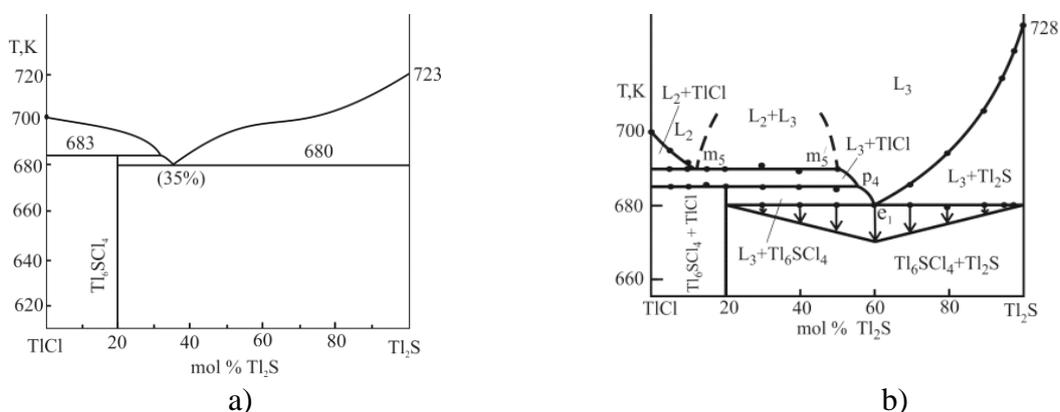


Fig 1. Phase diagram of the system TlCl-Tl<sub>2</sub>S: a) - [61], b) - [63]

**Tl-S-Br system.** TlBr-Tl<sub>2</sub>S and TlBr-S sections of this ternary system are quasi-binary [65,66]. According to [60], TlBr-Tl<sub>2</sub>S system is characterised by formation of incongruently melting ternary Tl<sub>6</sub>SBr<sub>4</sub> compound at 716K (Fig.2,a). The phase equilibrium in the system was re-examined; the formation of the above mentioned ternary compound proved, and coordinates of nonvariant equilibria specified (Fig. 2,b) [66].

TlBr-S system is of monotectic and degenerated eutectic type. Tl – S – Br ternary system was investigated at the Tl-TlBr-S concentration range, some isopleth sections and the projection of the liquidus surface of its phase diagram were constructed, types and coordinates of non- and monovariant equilibria determined [65,66].

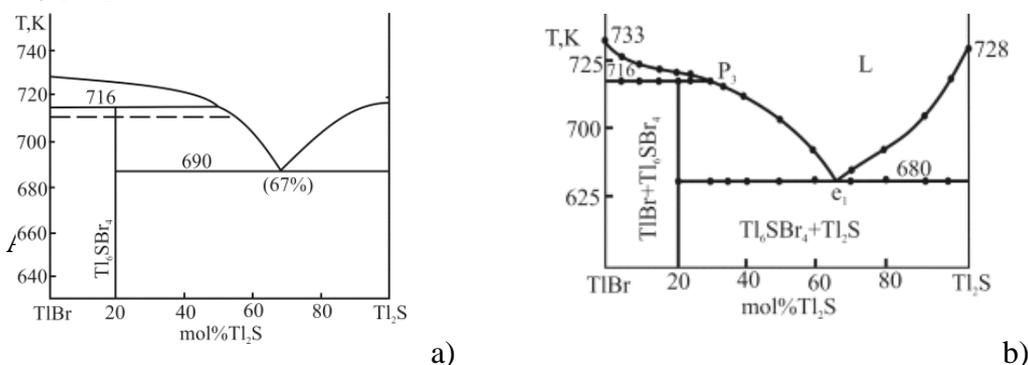
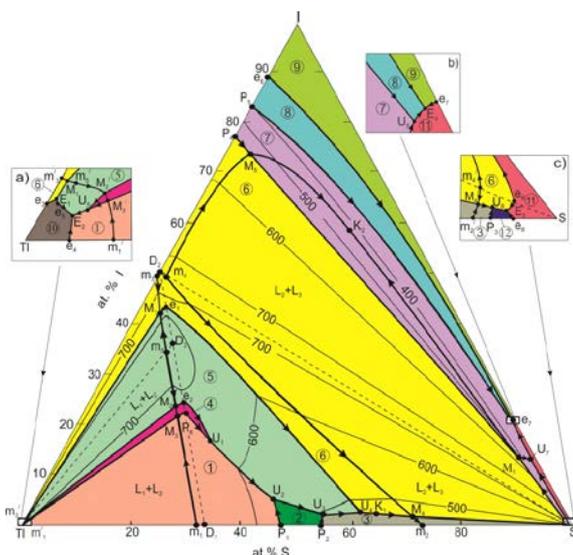


Fig.2. Phase diagram of the system TlBr-Tl<sub>2</sub>S: a) - [61], b) - [67]

**Tl – S – I system.** TII-Tl<sub>2</sub>S quasi-binary section is characterized by formation of ternary Tl<sub>6</sub>SI<sub>4</sub> compound melting congruently at 713K, and a phase of the Tl<sub>3</sub>SI composition that is stable at 640-670K temperature interval [61]. According to the results of [60], Tl<sub>6</sub>SI<sub>4</sub> melts congruently at 710K and has a ~4 mol % (according to [67,68] ~2 mol %) homogeneity area [69] report that, Tl<sub>6</sub>SI<sub>4</sub> melts congruently at 710K, Tl<sub>3</sub>SI – incongruently at 673K by peritectic reaction.

The latter decomposes at 653K according to solidphase reaction into Tl<sub>2</sub>S and Tl<sub>6</sub>SI<sub>4</sub> compounds.

The full description of phase equilibria of the Tl-S-I ternary system is given in the Fig.3 [65,69,70]. Three polythermic sections (TII-Tl<sub>2</sub>S, TII-S vñ Tl<sub>6</sub>SI<sub>4</sub>-Tl) of this system are quasi-binary. TII-S and Tl<sub>6</sub>SI<sub>4</sub>-Tl systems are characterized by monotectic and eutectic equilibria.



**Fig. 3.** The projection of the liquidus surface of the system Tl-TII-S. Primary crystallization areas: 1-Tl<sub>2</sub>S, 2-Tl<sub>4</sub>S<sub>3</sub>, 3-TIS, 4-Tl<sub>3</sub>SI, 5-Tl<sub>6</sub>SI<sub>4</sub> (γ), 6-(TII)<sub>II</sub>, 7-Tl<sub>2</sub>I<sub>3</sub>, 8-TII<sub>3</sub>, 9- I<sub>2</sub>, 10-Tl, 11-S and 12-Tl<sub>2</sub>S<sub>5</sub>

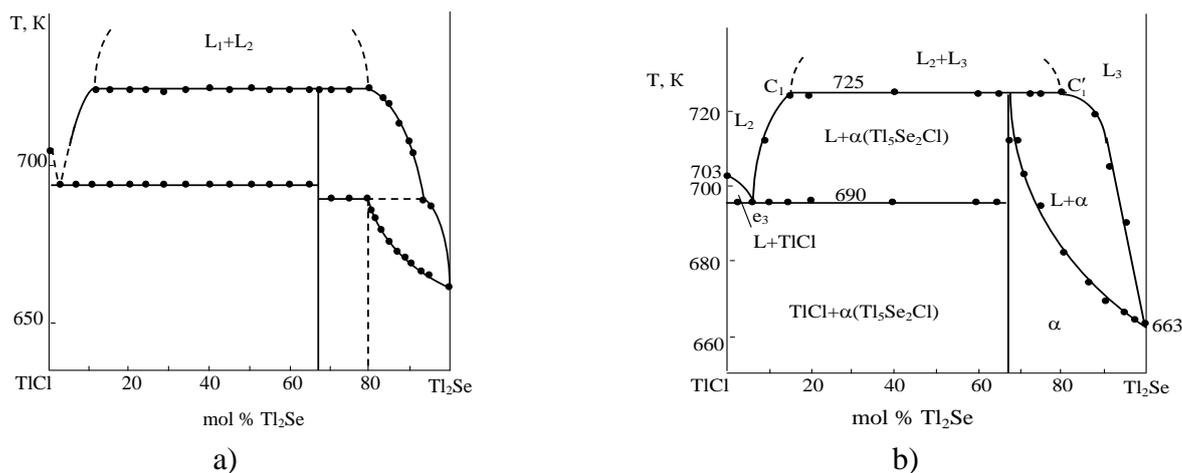
There are 2 immiscibility areas in the system Tl-S-I. Monovariant eutectic and peritectic equilibrium curves are transformed into nonvariant monotectic equilibria while

intersecting this immiscibility regions (Fig.3 - M<sub>1</sub>M<sub>1</sub>', M<sub>2</sub>M<sub>2</sub>', M<sub>3</sub>M<sub>3</sub>', M<sub>4</sub>M<sub>4</sub>', M<sub>5</sub>M<sub>5</sub>' conjugate pairs).

## 2.2. Phase equilibria in the Tl – Se – Cl (Br, I) systems

**Tl – Se – Cl system.** A ternary compound Tl<sub>5</sub>Se<sub>2</sub>Cl melting by sintectic reaction at 725K is formed in the first version of the Tl<sub>2</sub>Se-TlCl phase diagram [71] (Fig.4a). This compound is in eutectic equilibrium with TlCl, and in peritectic equilibrium with Tl<sub>2</sub>Se. The homogeneity area of Tl<sub>2</sub>Se at the peritectic reaction temperature is ~20 mol%. The T-x diagram of this system constructed by [72]

(Fig. 4,b) is consistent at 0-66.7 mol% Tl<sub>2</sub>Se composition interval with results of [71]. However, results obtained by [72,73] at the Tl<sub>5</sub>Se<sub>2</sub>Cl-Tl<sub>2</sub>Se composition range differs qualitatively from those given in [71] and is characterized by formation of continuous solid solutions. Note that this agrees with the fact that the primary constituents of the sub-system Tl<sub>5</sub>Se<sub>2</sub>Cl-Tl<sub>2</sub>Se are iso-structural (see section 3).



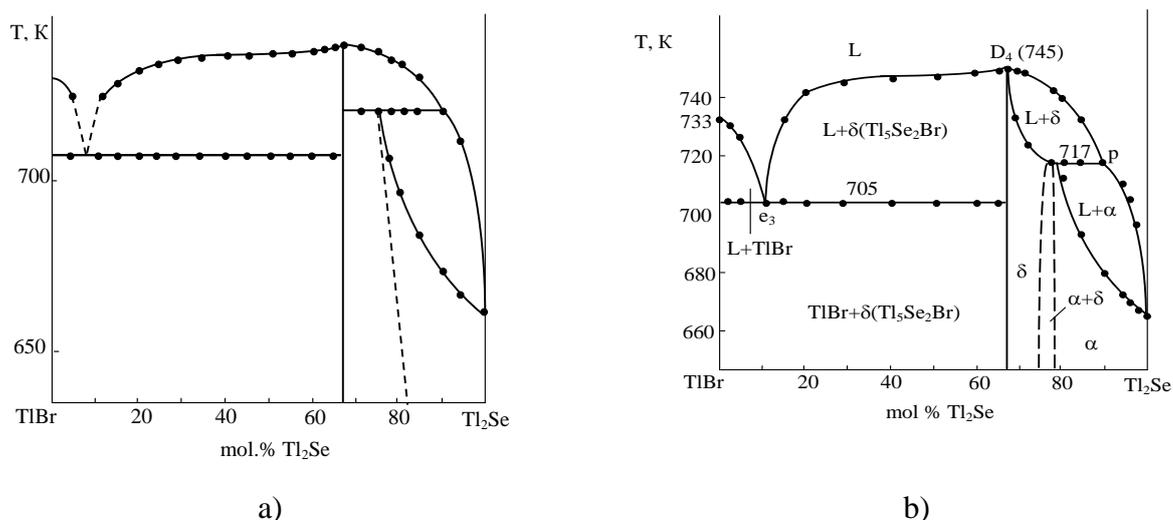
**Fig. 4.** Phase diagram of the system  $Tl_2Se-TlCl$ : a) – [72], b) – [73]

$TlSe-TlCl$  section is also quasi-binary and forms a phase diagram having monotectic and eutectic equilibria [60]. The  $Tl-S-Cl$  ternary system was studied in detail at the  $Tl-TlCl-Se$  concentration interval, and some isopleth sections, isothermal section at 400K, as well as the projection of the liquidus surface of its were constructed [72-76].

**Tl – Se – Br system.**  $TlBr-Tl_2Se$  system forms a congruently melting ternary compound  $Tl_5Se_2Br$  [71] (Fig.5a). It is stoichiometric to be in eutectic equilibrium with  $TlBr$  (~10 mol%  $Tl_2Se$ , 705K), and in peritectic equilibrium with  $Tl_2Se$  (~90 mol %  $Tl_2Se$ , 720K). There is a wide solid solution

area (75-100 mol%  $Tl_2Se$ , 720K) based on  $Tl_2Se$  [71]. New phase diagram [73,77] of the system  $TlBr-Tl_2Se$  coincides with the results of [71] at the  $TlBr-Tl_5Se_2Br$  composition interval, whereas it differs considerably from it at the  $Tl_5Se_2Br-Tl_2Se$  interval (Fig.5b). According to [71],  $Tl_5Se_2Br$  forms a large (66.7-75 mole%  $Tl_2Se$ ) solid solution area. The results of Ref. [71] agree with the results of Ref. [78-80] that  $Tl_5Se_2Br$  is of variable composition.

Experimental results in the  $Tl-TlBr-Se$  concentration range were summarized, some polythermal and isothermal sections, liquidus surface projection constructed in Ref. [77].



**Fig.5.** Phase diagram of the system  $Tl_2Se-TlBr$ : a) – [72], b) – [78]

**Tl – Se – I system.** The phase equilibria in the Tl–Se–I ternary system was investigated in a whole concentration range [60, 65, 71, 81-86]. Both ternary compounds -  $Tl_5Se_2I$  and  $Tl_6SeI_4$  detected in the Tl–Se–I system are formed on the TII– $Tl_2Se$  section [60, 71, 83-84].  $Tl_5Se_2I$  melts congruently at 720K,  $Tl_6SeI_4$  melts by peritectic decomposition at 705K [71].  $Tl_5Se_2I$  has a wide homogeneity region and is in peritectic equilibrium with  $Tl_2Se$ . However, according to the results of [60], both compounds are non-stoichiometric phases which melt congruently.

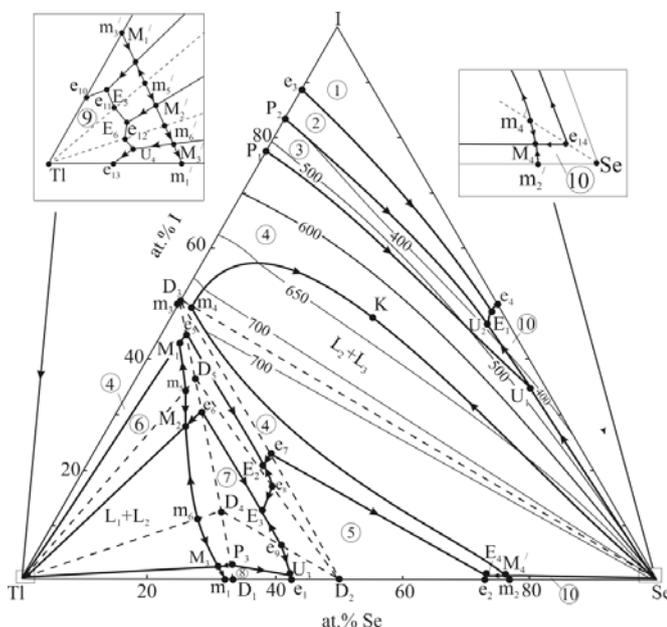
A re-study of the TII– $Tl_2Se$  system [83,84] confirmed the existence of congruently melting ternary compounds  $Tl_5Se_2I$  (720K) and  $Tl_6SeI_4$  (705K).

TlSe–TII [60], TlSe– $Tl_5Se_2I$  and TlSe– $Tl_6SeI_4$  [84,85] polythermal sections of the system Tl–Se–I are quasi-binary and form eutectic phase diagram. TII–Se [86],  $Tl_5Se_2I$ –Tl [84] and  $Tl_6SeI_4$ –Tl [84] polythermal sections are also quasi-binary and characterized by monotectic and degenerated eutectic equilibria.

The projection of the liquidus surface of the system Tl–Se–I is given by the Ref. [65,84].

It consists of 10 primary crystallization areas (Fig.6). Liquidus surfaces of the elemental Tl and Se are degenerated on the relevant vertices of a concentration triangle. The largest area of crystallization belongs to the high temperature modification of thallium monoiodide - (TII)<sub>II</sub>. This compound involves most non- and monovariant equilibria in the Tl–Se–I system.

Tl–Se–I system is divided into 8 sub-systems by 7 quasi-binary sections (Fig.6, dashed lines). 3 of those sub-systems (Tl–TII– $Tl_6SeI_4$ , Tl– $Tl_6SeI_4$ – $Tl_5Se_2I$ , TII–TlSe–Se) are characterized by nonvariant monotectic and eutectic equilibria, two of them (TII– $Tl_6SeI_4$ –TlSe,  $Tl_6SeI_4$ – $Tl_5Se_2I$ –TlSe) are of eutectic type, one -  $Tl_2Se$ – $Tl_5Se_2I$ –TlSe is of transition type, and one - Tl– $Tl_2Se$ – $Tl_5Se_2I$  is of monotectic and transition types. The latter system - TII–Se–I covers half of the concentration triangle and reflects a series of non- and monovariant equilibria. The homogeneity areas and dissociation in the liquid state of the  $Tl_5Se_2I$  and  $Tl_6SeI_4$  compounds are comprehensively studied [61,67,68,78-80].



**Fig.6.** The projection of the liquidus surface of the system Tl–Se–I. Primary crystallization areas: 1– $I_2$ , 2– $TlI_3$ , 3– $Tl_2I_3$ , 4–(TII)<sub>II</sub>, 5–TlSe, 6– $Tl_6SeI_4$ , 7– $Tl_5Se_2I$  ( $\delta$ -phase), 8– $Tl_2Se$  ( $\alpha$ -phase), 9–Tl, 10–Se.

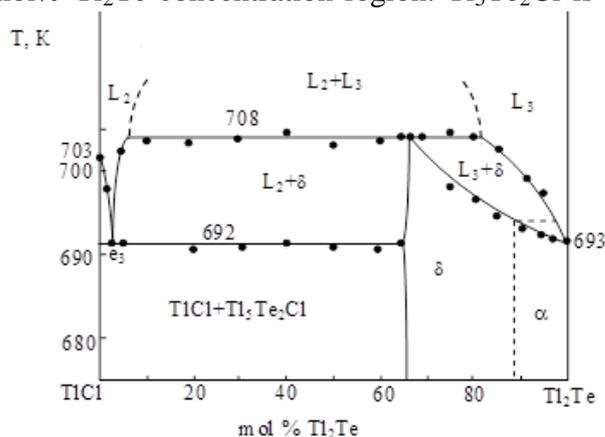
### 2.3. Phase equilibria in the Tl – Te – Cl (Br,I) systems

**Tl – Te – Cl system.** Phase equilibria in this system were studied by TlCl–Te( $TlTe$ ,  $Tl_2Te_3$ ),  $Tl_5Te_2Cl$ –

$Tl(TlTe, Tl_2Te_3, Tl_5Te_3)$  polythermal sections [73,76,87-89]. The further study of phase equilibria in the Tl–TlCl–Te composition area

was carried out by [90-92], a number of polythermal sections, the projection of a liquidus surface of the system constructed, types and coordinates of non- and monovariant equilibria determined.

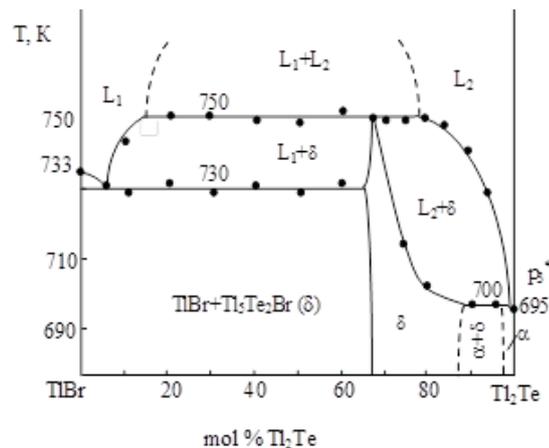
TlCl-Tl<sub>2</sub>Te quasi-binary section is characterized by formation of ternary Tl<sub>5</sub>Te<sub>2</sub>Cl that melts incongruently by syntectic reaction at 708 K [90,92]. The aliquidation area at this temperature ranges between ~7-85 mol% Tl<sub>2</sub>Te concentration region. Tl<sub>5</sub>Te<sub>2</sub>Cl is



a)

in eutectic equilibrium with TlCl, whereas it is a solid solution area with Tl<sub>2</sub>Te (Fig. 7,a).

The main feature of the system Tl-TlCl-Te is the existence of a wide immiscibility areas on the basis of 3 different liquid phases. These aliquidation fields cover more than 90% of the Tl-TlCl-Te concentration triangle. There is a wide triple immiscibility region which is surrounded by corresponding double immiscibility areas [92].



b)

**Fig. 7.** Phase diagram of the systems TlCl-Tl<sub>2</sub>Te (a) [94] and TlBr-Tl<sub>2</sub>Te (b) [97]

**Tl – Te – Br system.** This system was studied in the Tl-TlBr-Te composition area [87, 89, 91, 93-95]. It revealed that the only quasi-binary section of this system - TlBr-Tl<sub>2</sub>Te forms a phase diagram with syntectic, eutectic and peritectic equilibria. Tl<sub>5</sub>Te<sub>2</sub>Br melts by syntectic reaction at 750K. Its homogeneity area occupies 66.7-87 mol% Tl<sub>2</sub>Te concentration interval [91,95] (Fig. 7,b).

TlBr-Te system is quasi-binary, whereas TlBr-TlTe(Tl<sub>2</sub>Te<sub>3</sub>), Tl<sub>5</sub>Te<sub>2</sub>Br-TlTe, Tl<sub>5</sub>Te<sub>2</sub>Br-Tl<sub>5</sub>Te<sub>3</sub> isopleth sections are stable in subsolidus. The latter forms a continuous series of solid solutions.

The projection of the liquidus surface consists of 5 fields which agrees with the primary crystallization of the TlBr,  $\delta$ -phase, TlTe, Tl<sub>2</sub>Te<sub>3</sub> and elemental Te [95]. Formation of the triple immiscibility area in the system is due to the presence of extensive double immiscibility areas on all sides of the Tl-TlBr-Tl<sub>2</sub>Te subsystem. The triple immiscibility area is characterized by the nonvariant syntectic  $L_1+L_2+L_3 \rightarrow \delta$  reaction.

In Ref. [96-102] ternary compounds Tl<sub>2</sub>TeBr(I)<sub>6</sub> were detected, their homogeneity areas defined and some properties of their monocrystals examined.

**Tl – Te – I system** was investigated in a whole concentration interval.

TlI-Tl<sub>2</sub>Te quasi-binary section forms a ternary Tl<sub>5</sub>Te<sub>2</sub>I compound meltig at 775K by syntectic reaction [60]. Tl<sub>5</sub>Te<sub>2</sub>I is a phase of variable composition and has a ~10 mol % homogeneity region along the TlI-Tl<sub>2</sub>Te system (Fig.8).

Tl<sub>2</sub>TeI<sub>6</sub> compound melting congruently at 700K was detected on the TlI-TeI<sub>4</sub> quasi-binary section, its crystal structure and some physical properties were studied [97-102]. New option of the TlI-TeI<sub>4</sub> phase diagram [108] differs from the one described in Ref. [100] by melting temperature of Tl<sub>2</sub>TeI<sub>6</sub> - 654K.

T-x-y digram of the Tl-Te-I system is given in [105-110]. It found that TlI-Te, TlI-Tl<sub>2</sub>Te, TlI-TeI<sub>4</sub>, Tl<sub>2</sub>TeI<sub>6</sub>-Te and Tl<sub>2</sub>TeI<sub>6</sub>-I polythermal sections are quasi-binary. The

phase diagram of the TlI-Tl<sub>2</sub>Te system has been specified by [98]. The results obtained are close to [60].

The solid-phase diagram of the Tl-Te-I system at 300K reflects 2 ternary compounds

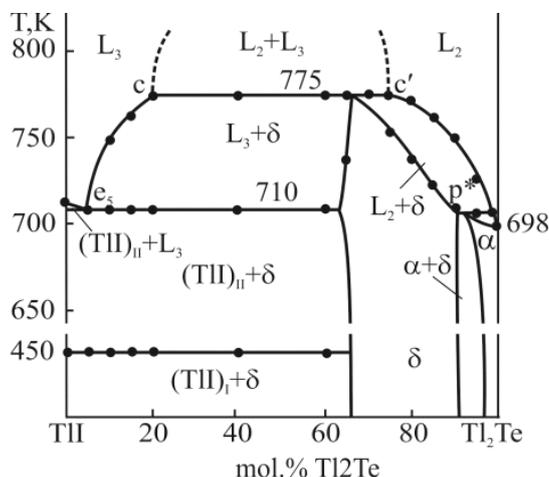


Fig. 8. Phase diagram of the system TlI-Tl<sub>2</sub>Te

Projection of the liquidus surface consists of 11 primary crystallization areas (Fig.9) [108]. Liquidus surfaces of the elemental Tl and  $\alpha$ -phase based on Tl<sub>2</sub>Te are degenerated.

To sum up, let's look at characteristic features of phase equilibria in the Tl-X-Hal systems.

6 ternary compounds of the type Tl<sub>5</sub>X<sub>2</sub>Hal (in all Se and Te containing), 4 compounds of the type Tl<sub>6</sub>XHal<sub>4</sub> (in all S containing systems) and Tl<sub>2</sub>TeHal<sub>6</sub> compounds have been detected in the Tl-X-Hal systems. It revealed that all compounds, except for the

latter ones, are formed on the TlHal-Tl<sub>2</sub>X quasi-binary sections. Tl<sub>5</sub>X<sub>2</sub>Hal compounds decompose incongruently by syntectic reaction except for the Tl<sub>5</sub>Se<sub>2</sub>I and Tl<sub>5</sub>Se<sub>2</sub>Br that melt congruently. Tl<sub>6</sub>XHal<sub>4</sub> compounds melt congruently (Tl<sub>6</sub>SI<sub>4</sub> and Tl<sub>6</sub>SeI<sub>4</sub>) or by peritectic decomposition.

Tl<sub>5</sub>X<sub>2</sub>Hal compounds are phases of variable composition possessing a wide homogeneity area, whereas Tl<sub>6</sub>XHal<sub>4</sub> and Tl<sub>2</sub>TeHal<sub>6</sub> compounds are practically stoichiometric. Crystal structures of all three types of compounds are discussed in the next section.

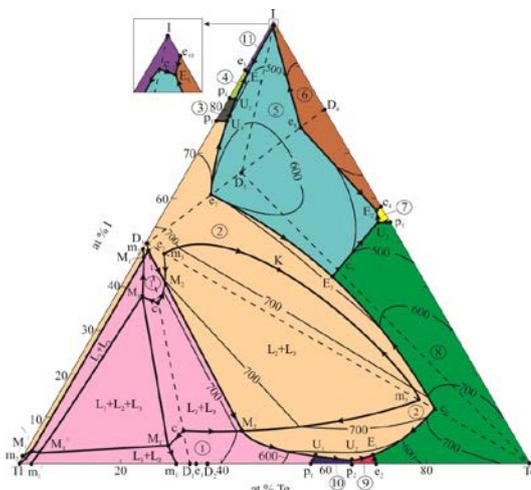


Fig.9. Liquidus surface projection of the system Tl-Te-I. Primary crystallization fields:

- 1-  $\delta$  phase, 2- (TlI)<sub>II</sub>, 3-Tl<sub>2</sub>I<sub>3</sub>,
- 4-TlI<sub>3</sub>, 5-Tl<sub>5</sub>Te<sub>2</sub>I, 6-TeI<sub>4</sub>, 7-TeI, 8-
- Te, 9-Tl<sub>2</sub>Te<sub>3</sub>, 10- TlTe, 11- I<sub>2</sub>

The characteristic feature of the Tl-X-Hal systems is the existence of wide immiscibility areas in a liquid state. There are

triple immiscibility areas together with double ones in the 6 systems out of the 9 Tl-X-Hal systems. The comparative analysis of the

liquidus surfaces shows that the character phase of equilibria changes regularly in three directions of  $S \rightarrow Se \rightarrow Te$  and  $Cl \rightarrow Br \rightarrow I$ , which is due to changes in the nature of chemical bonding in these systems.

Criteria of aliquidation was revealed in the Tl-X-Hal ternary systems as a result of the comparative analysis of T-x diagrams of more than 50 quasi-binary and quasi-stable sections [109, 110]. It showed that there is an immiscibility area in the systems when the

degree of ionisation (DOI) is more than 17, and the systems with  $DOI < 16-17$  don't demonstrate aliquidation.

To summarize, there is a rare case of practice in the  $Tl_5X_2Hal-Tl_2X$  and  $Tl_5Te_2Hal-Tl_5Te_3$  sections of the Tl-X-Hal systems. Despite the presence of extensive immiscibility areas in liquid state, these systems are characterized by formation a wide or continuous solid solution areas in subsolidus.

### 3. Crystal structure and some physico-chemical properties of thallium chalcogenides

The literature data on the crystal structure and some physico-chemical properties of thallium chalcogenides are discussed in this section. As mentioned above (section 2), there are 3 types of ternary compounds in Tl-X-Hal systems:  $Tl_5X_2Hal$ ,  $Tl_6XHal_4$  and  $Tl_2TeHal_6$ . Crystal lattice types and parameters of those compounds are listed in the Table 1.

$Tl_5X_2Hal$  compounds are crystallized in a tetragonal lattice of the  $Tl_5Te_3$  type. The crystal structure of  $Tl_5Te_3$  has been analyzed in a number of studies [111-114] following which this compound is crystallized in a tetragonal structure of the  $Cr_5B_3$  type (Sp.gr.  $I4/mcm$ ). In the unit cell there are 4 formula units ( $z = 4$ ).

The main structural elements of the  $Tl_5Te_3$  crystal lattice (Fig. 10) are tellurium octahedra in which tellurium atoms have two different positions: part (Te1) is located at two opposite vertices of octahedra along the  $c$  axis, and the other part (Te2) occupies the remaining positions.

Thallium atoms are also subdivided into 2 types according to their positions in the crystal lattice: one part of the cations - Tl (1) is located in equivalent positions with a multiplicity of 16, and another part - Tl (2) - in positions with a multiplicity of 4. Chemical composition of the unit cell is  $Tl_{16}Tl_4Te_8Te_4$ . Considering the electroneutrality condition, it can be assumed that in positions with a multiplicity of 4, the  $Tl^+$  and  $Tl^{3+}$  cations alternate and the unit cell can be represented as  $Tl_{16}[(Tl_{0,5}^{1+}B_{0,5}^{3+})Te_3]_4$ . Linking to the vertices, the octahedra form a skeleton of the com-

position  $Tl_4Te_{12}$  or  $(TlTe_3)_4$  (Fig. 10, a). Thallium prisms are formed around the octahedra, and antiprisms of thallium are formed around the anions at the vertices of the octahedra, which bind the octahedra along the  $c$  axis. Prisms and anti-prisms are alternated along the  $c$  axis. These structural elements create a structure of the  $Tl_5Te_3$  type with the chemical composition of the cell  $Tl_{16}(TlTe_3)_4$  (Fig. 10, b).

Due to the above mentioned features of the crystal lattice,  $Tl_5Te_3$  forms a series of cation- and anion-substituted derivatives. Typical representatives of cation-substituted ternary analogues of  $Tl_5Te_3$  are compounds of the types  $Tl_9BTe_6$  (B-Sb, Bi, In, Au, rare earth elements) [115-119] and  $Tl_4ATE_3$  (A-Sn, Pb, Cu, Mo, Nd) [115,118, 120-122].

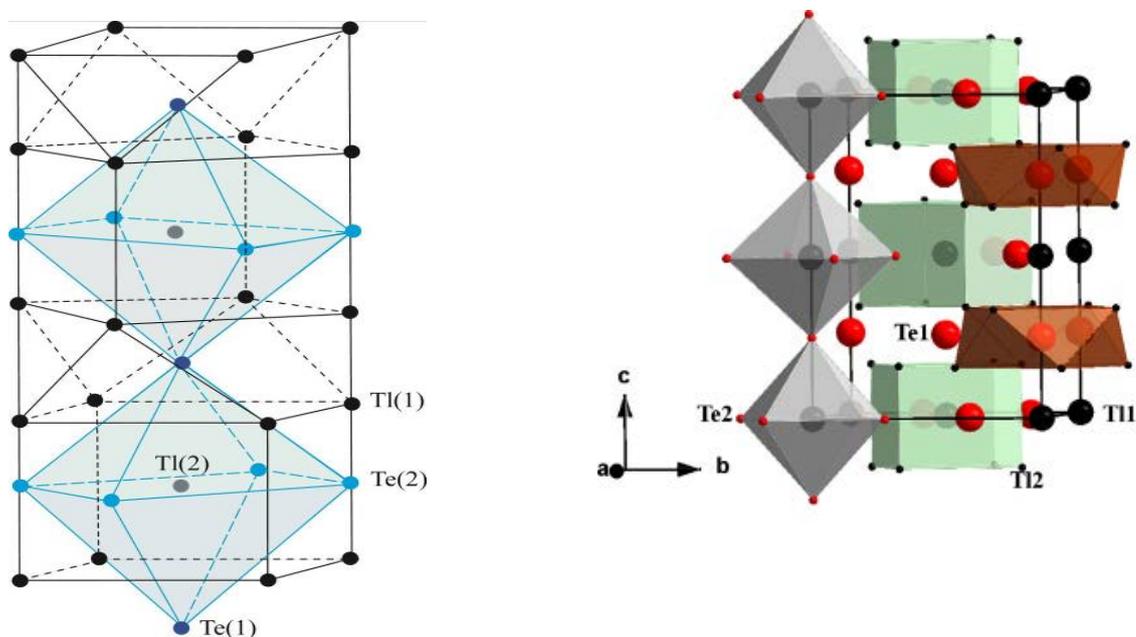
$Tl_4ATE_3$  and  $Tl_{16}[A^{2+}Te_3]_4$  compounds are formed when all the thallium atoms in the centers of the octahedra (Tl2) are replaced by the  $A^{2+}$  cations. Replacement of half of the Tl atoms located in the vacancies of octahedra (Tl2), with  $B^{3+}$  cations leads to the formation of  $Tl_9BTe_6$  compounds [117].

The crystal structures of  $Tl_5X_2Hal$  compounds - anion-substituted derivatives of  $Tl_5Te_3$  were studied in detail. According to the available data (Table 1), all these compounds, with an exception of  $Tl_5Se_2Cl$  (Sp.gr  $P4/ncc$ ), belong to the spatial group  $I4/mcm$ . Se or Te atoms located at two opposite vertices of the octahedra are replaced by halogen atoms in the crystal lattices of these compounds.

The interesting aspect of the anion replacement is that it enables formation of solid solutions at the composition interval  $Tl_5Te_{3-x}Hal_x$  ( $0 < x \leq 1$ ). Meantime, Tl gradually

changes its oxidation state by switching to  $Tl^+$  and provides electroneutrality. The analysis of the  $Tl_5Te_3$  type structures show that both  $Tl_5Te_3$  and  $Tl_5Se_{3-x}$  compounds, as well as

their structural analogues are phases of variable composition with a wide one-sided homogeneity areas.



**Fig.10.** Crystal structure of  $Tl_5Te_3$ . The main structural element (a), the projection along the surfaces  $b$ ,  $c$  (b)

**Table 1.** Crystal lattice parameters of  $Tl_5Te_3$  and thallium chalcogenides

Compound	Syngony, Space Group	Lattice parameters, Å	Reference
$Tl_5Te_3$	Tetragonal, $I4/mcm$	$a=8.930; c=12.598; z=4$	[114]
$Tl_6SI_4$	Tetragonal, $P4/mnc$	$a=9.176; c=9.608; z=2$	[61,82]
$Tl_6SBr_4$	“—“	$a=8.721; c=9.328; z=2$	[61,82]
$Tl_6SCl_4$	“—“	$a=8.830; c=9.170; z=2$	[61,82]
$Tl_6SeI_4$	“—“	$a=9.178; c=9.675; z=2$	[81,82]
$Tl_5Se_2I$	Tetragonal, $I4/mcm$	$a=8.663; c=13.463; z=4$	[71]
$Tl_5Se_2Br$	“—“	$a=8.594; c=12.788; z=4$	[123]
$Tl_5Se_2Cl$	Tetragonal, $P4/ncc$	$a=8.565; c=12.741; z=4$	[123]
$Tl_2TeI_6$	Monoclinic, $P2_1/c$	$a=7.765; b=8.174; c=13.756;$ $\beta=124.2^\circ; z=2$	[102]
$Tl_2TeBr_6$	Tetragonal, $P4/mnc$	$a=7.468; c=10.682; z=2$	[102]
$Tl_2TeCl_6$	Cubic, $Fm-3m$	$a=10.107; z=4$	[124]
$Tl_5Te_2I$	Tetragonal, $I4/mcm$	$a=9.001; c=13.291; z=4$	[108]
$Tl_5Te_2Br$	Tetragonal, $I4/mcm$	$a=8.974; c=12.812; z=4$	[125]
$Tl_5Te_2Cl$	Tetragonal, $I4/mcm$	$a=8.921; c=12.692; z=4$	[90,92]

Crystal structure of  $Tl_5Se_2Hal$  compounds was studied, their lattice parameters, and positions of atoms in the unit cell determined [71,123]. All of them are crys

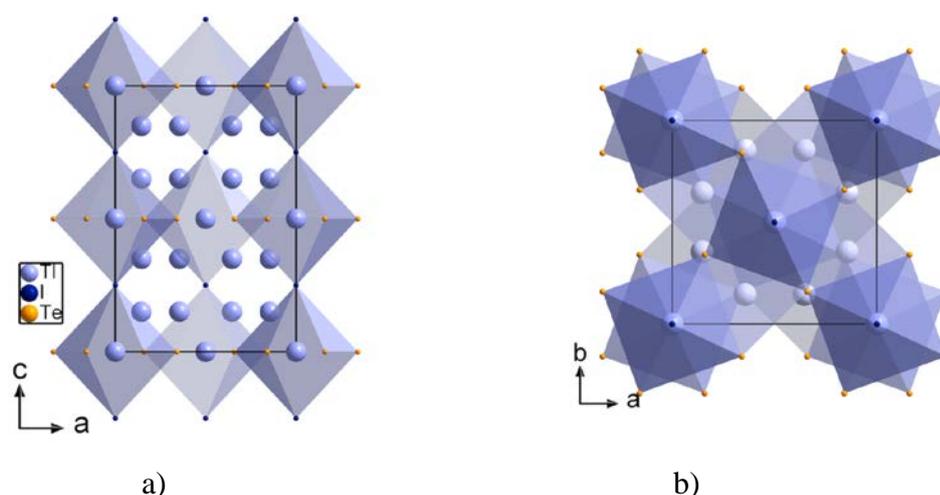
tallized with the  $In_5Bi_3$  structure which is a subfamily of the  $Cr_5B_3$  type:  $Tl_5Se_2I(Br)$  is crystallized in the space group  $I4/mcm$ ,  $Tl_5Se_2Cl$  as a lower symmetric distorted vari-

ant in  $P4/ncc$ . Around one central thallium atom there are slightly distorted octahedra of two halogen and four selenium atoms occur. These octahedra are covered by cubes of thallium atoms. The octahedra are interconnected via common vertices and the cubes via common edges to give the three-dimensional structure [123].

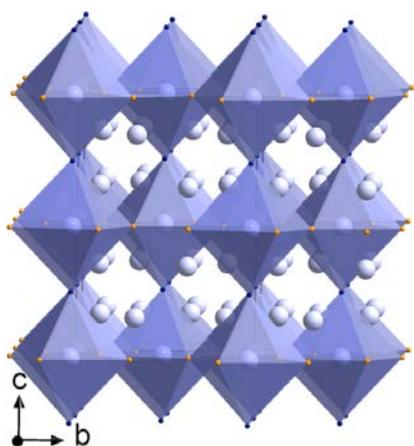
The crystal structure study of  $Tl_5Te_2I(Cl/Br)$  compounds was carried out by the Rietveld method based on powder X-rays analysis. The unit cell parameters and isotropic atomic positions of the atoms were refined by TOPAS-4.2 program [90,125].

Based on the phase diagram of the Tl-Te-I system, it was proposed and implemented

an original method of growing single crystals of  $Tl_5Te_2I$  which melts incongruently by syntectic reaction [65,126]. The method has been used in order to control the crystallization process from two immiscible liquid phases  $L_1 + L_2$ . In this method the liquid phase  $L_1$  was in dynamic equilibrium with another liquid phase and the compositions of the coexisting liquid phases always constant at the given temperature. During very slow crystallization from the  $L_1$ , the phase  $L_2$  gradually dissolves in  $L_1$ , and provides constancy of its composition to form consequently the  $Tl_5Te_2I$  by the syntectic reaction at 775 K according to phase diagram.

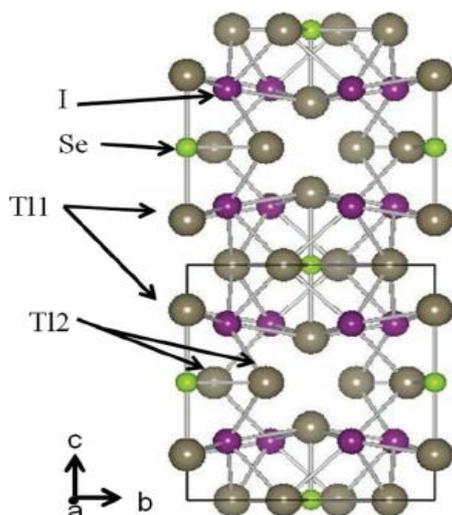


**Fig.11.** Crystal structure of  $Tl_5Te_2I$ : a) view along the [010] direction; b) Projection along the  $c$ -axis. Tl atoms in lighter color.

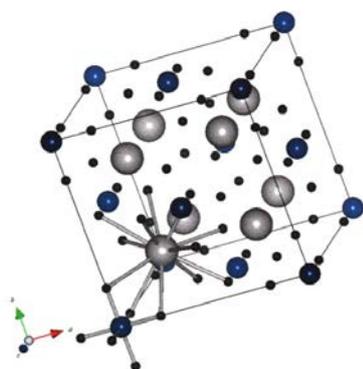


**Fig. 12.** Perspective view of  $Tl_5Te_2I$  showing a layer of  $Tl_5Te_2I$  corner-sharing octahedral. Color codes like in Fig. 11 and Tl atoms in lighter color

Single crystal X-ray diffraction data was collected at room temperature on a STOE StadiVari diffractometer. The structure was refined using as starting model that of  $Tl_5Te_3$ ; refinement performed with the program system JANA2006. All atoms were refined with occupancies according to the stoichiometric formula, anisotropic thermal displacement parameters and without any constrain. Images of chemical structure were visualized using VESTA software. Full crystal structure details, data collection and structure refinement details are given in [108]. The structure can be described as layers of corner sharing  $TlTe_4I_2$  octahedral ( $Tl_2$  central atom) with additional interstitial  $Tl_1$  atoms with trigonal prism coordination. Octahedra consecutive along the  $c$ -axis are rotated 45° relative to each other. A projec-



**Fig. 13.** The crystal structure of  $Tl_6SeI_4$ . Two distinct sites for Tl ions are shown as Tl1 and Tl2, respectively [50].



**Fig.14.** The unit cell of  $Tl_2TeBr(I)_6$  compounds. Tl cations are shown by a large gray circles, Te atoms – by dark blue circles and the halogen atoms – by small circles.

The synthesis, identification of the homogeneity areas, growth of monocrystals and study of some properties of the  $Tl_2TeHal_6$  (Hal – I,Br) compounds are reported in [96-102,128-130]. The crystal structure of  $Tl_2TeBr_6(I_6)$  compounds was studied in Ref. [98,102] (Fig.14). Both compounds are crys-

talized along the [010] and [001] directions, and perspective view of the structure are shown in the Fig. 11 and 12 [127] reported formation mechanism of the anion- and cation substituted analogs of  $Tl_5Te_3$  takes into consideration structural properties of this compound and the electronic structure of thallium. The rational content of these analogues  $Tl_4BX_3$ ,  $Tl_9BX_6$  and  $Tl_5X_2Hal$  was justified, and the reasons for their one-sided stoichiometry determined.

$Tl_6SHal_4$  and  $Tl_6SeI_4$  (Hal– I,Br,Cl) ternary compounds are crystallized in a tetragonal lattice of the type  $Tl_4HgBr_6$  [61,81,82] (Fig.13). There are two distinct crystallographic sites for Tl cations. Each  $Tl^+$  ion is coordinated with four  $Hal^-$  ions and one  $(S^{2-})Se^{2-}$  ion.

tallized in a crystal lattice of the  $K_2PtCl_6$  type.  $Tl_2TeCl_6$  ternary compound crystallize in a cubic unit cell [124] (Table 1).

The methods for the synthesis and monocrystal growth, and some physico-chemical properties of the  $Tl_6XHal_4$  and

Tl<sub>5</sub>Se<sub>2</sub>Hal chalcogenides were analyzed in Ref. [66,67,78,131,132].

The 3D homogeneity areas were defined and some electrophysical properties of Tl<sub>6</sub>SI<sub>4</sub>

and Tl<sub>6</sub>SeI<sub>4</sub> monocystals examined in [131,132].

#### 4. Thermodynamic properties of thallium chalcogenides and related phases

The fundamental thermodynamic characteristics of substances are important to ensure the effectiveness of thermodynamic calculations for optimization of various processes, in particular, conditions for the synthesis and growth of crystals [133, 134].

Thermodynamic properties of intermediate phases in the Tl-X-Hal systems were examined through the EMF measurements of the concentration chains

(-) Tl (solid) | gluserol + KCl + TlCl | (Tl-X-Hal) (solid) (+) (1)

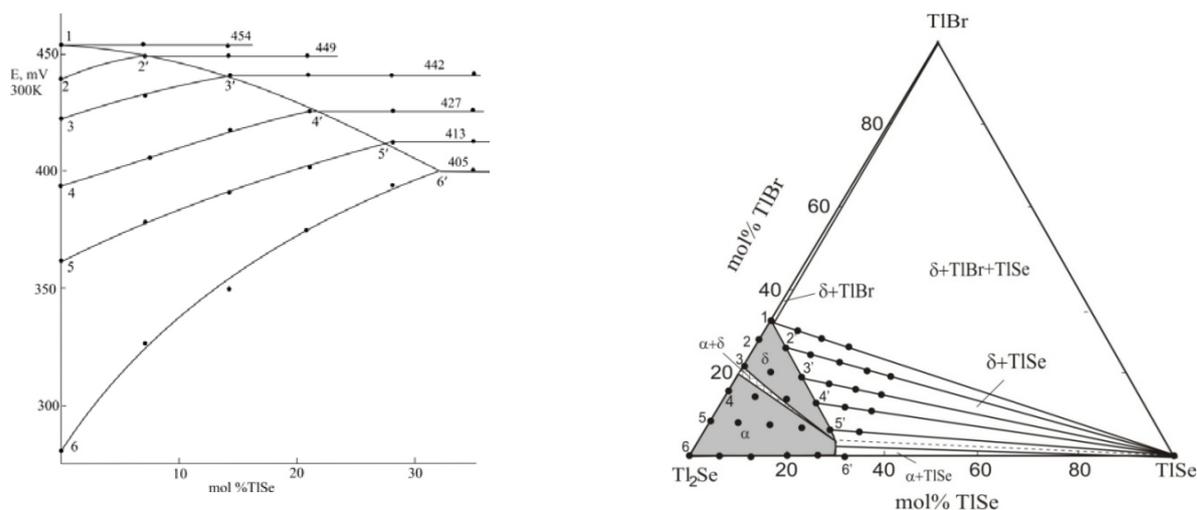
In the chains of type (1), metallic thallium was used as the left electrode, while equilibrium alloys of the Tl-X-Hal systems were used as the right electrodes. A saturated glycerin solution of KCl with the addition of 0.1 mass% TlCl was used as an electrolyte. EMF was measured by the compensation method in the temperature range of 300–430 K and in the 300-380K temperature intervals for sulphur containing systems.

**Table 2.** Thermodynamic properties of thallium chalcogenides

Compound	$-\Delta_f G^0(298K)$	$-\Delta_f H^0(298K)$	$S^0(298K),$ (J·mol <sup>-1</sup> ·K <sup>-1</sup> )	Reference
	(kJ·mol <sup>-1</sup> )			
Tl <sub>6</sub> SI <sub>4</sub>	601.7±2.5	595.1±4.0	672±10	[69]
Tl <sub>6</sub> SBr <sub>4</sub>	767.0±2.9	790.3±5.2	644±9	[65]
Tl <sub>6</sub> SCl <sub>4</sub>	833.5±3.7	928.1±14.0	599±9	[64,65]
Tl <sub>6</sub> SeI <sub>4</sub>	613.1±1.5	609.7±2.6	671±5	[84]
Tl <sub>5</sub> Se <sub>2</sub> I	341.7±0.8	345.3±2.5	449±8	[84]
Tl <sub>5</sub> Se <sub>2</sub> Br	374.3±1.0	384.3±2.7	447.6±6.4	[77]
Tl <sub>5</sub> Se <sub>2</sub> Cl	392.8±1.1	421.6±5.1	433.9±7.2	[72]
Tl <sub>5</sub> Te <sub>2</sub> I	300.4±1.3	301.1±2.3	475.8±6.6	[104]
Tl <sub>5</sub> Te <sub>2</sub> Br	340.6±1.6	344.5±2.7	483.4±6.2	[95]
Tl <sub>5</sub> Te <sub>2</sub> Cl	355.9±1.1	377.1±5.0	474.1±6.8	[92]

The homogeneity areas of phases were refined, the relative partial molar functions of thallium in alloys and the integral thermodynamic functions of thallium chalcogenides and solid solutions based on them calculated on the basis of the results of EMF measurements of the chains type (1). The standard thermodynamic functions of formation and the standard entropies of ternary thallic chalcogenides are summarized in the Table 2.

The results of EMF measurements devoted to the determination of homogeneity areas of the Tl<sub>5</sub>X<sub>2</sub>Hal compounds and calculation of thermodynamic functions of solid solutions are presented and discussed in [84,104,135-138]. As an example, Fig. 15 presents the data of [138] on the refinement of the homogeneity region of the Tl<sub>5</sub>Se<sub>2</sub>Br compound.



**Fig.15.** Application of the EMF method for the determination of homogeneity area of  $Tl_5Se_2Br$  compound [138]. a) the concentration dependence of the EMF of chains type (1) for alloys along the ray lines from the vertex TlSe b) a fragment of the phase diagram of the Tl-Se-Br system at 300 K.

The relative partial thermodynamic functions of thallium in the  $Tl_5Te_2Br_xI_{1-x}$  solid solutions at 298K were calculated and their dependence graph upon concentration constructed. It has been established that formation of solid solutions in the  $Tl_5Te_2Br-Tl_5Te_2I$  system is not accompanied by significant energy and structural changes, and the formed  $\delta$ -phase is a solid solution of substitution type [139].

The standard thermodynamic functions of formation and atomization, as well as standard entropies of thallium chalcocahalides and variable phases on their base were

calculated on the basis of partial thermodynamic functions of thallium using phase diagrams [140].

By using EMF measurements a regulated complex of thermodynamic properties for the  $Tl_6SBr_4$  compound was obtained. The data regarding the thermodynamic properties of all thallium chalc-halides were systematized and comparatively analyzed. Some correlations between thermodynamic functions of thallium chalc-halides and their binary constituents, as well as the ionization degree of chemical bonding were revealed [141,142].

## 5. Phase equilibria and physico-chemical properties of intermediate phases in some multi-component systems containing thallium, chalcogen and halogen

Some concentration areas of the Tl-  $X-X'$ -Hal and Tl- $X$ -Hal-Hal' ( $X, X'$ - chalcogen; Hal, Hal'- halogen) four-component systems are of a significant interest in terms of search and development of scientific basis for designing new multicomponent phases based on thallium chalcocahalides. Another way for varying the properties of ternary compounds is

to investigate the phase equilibrium in systems composed of them and their structural analogues. In this regard,  $Tl_5Te_2Hal-Tl_9B^VX_6$  and  $Tl_5Te_3-Tl_5Te_2Hal-Tl_9B^VX_6$  systems are very interesting, since all initial constituents of these systems are isostructural thermoelectrics possessing a low thermal conductivity [29].

### 5.1. Tl–X–X'–Hal (X, X'–chalcogen, Hal–halogen) systems.

The phase equilibria in the systems Tl<sub>2</sub>S–Tl<sub>2</sub>Se–TlI(Br) were studied, some polythermal sections and the liquidus surface projections of these systems constructed [143,144]. It was shown that, the Tl<sub>6</sub>SI<sub>4</sub>–Tl<sub>6</sub>SeI<sub>4</sub> [145,146] and Tl<sub>2</sub>S–Tl<sub>5</sub>Se<sub>2</sub>I [147] quasi-binary sections divide the Tl<sub>2</sub>S–Tl<sub>2</sub>Se–TlI system into 3 sub-systems. The Tl<sub>2</sub>S–Tl<sub>2</sub>Se–Tl<sub>5</sub>Se<sub>2</sub>I subsystem is of monotectic eutectic type [143] due to the formation of continuous solid solutions along the Tl<sub>6</sub>SI<sub>4</sub>–Tl<sub>6</sub>SeI<sub>4</sub> section [145].

3 monocrystals of the Tl<sub>6</sub>SI<sub>4</sub>–Tl<sub>6</sub>SeI<sub>4</sub> solid solutions were grown by direct crystallization and their some physico-chemical properties studied [148].

The physicochemical interaction of components in the TlI–Tl<sub>6</sub>SI<sub>4</sub>–Tl<sub>6</sub>SeI<sub>4</sub> system was studied by means of DTA and X-ray analysis, the projection of the liquidus surface and the TlI–Tl<sub>6</sub>S<sub>0,5</sub>Se<sub>0,5</sub>I<sub>4</sub> polythermal section of the system were constructed [149]. The monovariant eutectic L→α+β' equilibrium curve of the TlI–Tl<sub>6</sub>S<sub>0,5</sub>Se<sub>0,5</sub>I<sub>4</sub> system is calculated based on mathematical modeling.

Three vertical cross sections (Tl<sub>2</sub>S–Tl<sub>2</sub>Se, Tl<sub>2</sub>S–Tl<sub>5</sub>Se<sub>2</sub>Br, Tl<sub>6</sub>SBr<sub>4</sub>–Tl<sub>5</sub>Se<sub>2</sub>Br) and the liquidus surface projection of the Tl<sub>2</sub>S–Tl<sub>5</sub>Se<sub>2</sub>Br–TlBr system were constructed, the boundaries of phase areas, as well as the coordinates of nonvariant equilibria determined [150].

### 5.2. Tl–X–Hal–Hal' systems

The research results of the systems composed of selenides and halides of thallium are reported in [151-153]. It revealed that the TlSe–TlCl–TlI system belongs to the nonvariant eutectic type [151], whereas TlSe–TlBr–TlI [152] and TlSe–TlCl–TlBr [153] are of the monovariant eutectic type.

Tl<sub>5</sub>Se<sub>2</sub>Br–Tl<sub>5</sub>Se<sub>2</sub>I and Tl<sub>5</sub>Te<sub>2</sub>Br–Tl<sub>5</sub>Te<sub>2</sub>I systems are characterized by formation of continuous solid solutions [154,155].

The interaction between A<sub>2</sub>TeC<sub>6</sub> (A–K, Rb, Cs, Tl; C–Br, I) compounds was studied and their T-x diagrams constructed [99,154]. These systems are quasi binary and form a phase diagram of eutectic type containing a large solid solution areas on the base of primary constituents.

The effect of the crystallochemical factor on the interaction of the components in these systems was determined by the ratio of the type of A<sub>2</sub>TeC<sub>6</sub> compounds and the extent of the interaction of these cations and their anions in determining the nature of the interaction in these systems.

By the influence of the crystallochemical factor on the interaction of components in these systems it was determined that the main parameter defining the structure type of A<sub>2</sub>TeC<sub>6</sub> compounds and the character of chemical

interaction in these systems is the size ratio of cations and anions.

The phase equilibria in the T<sub>2</sub>Te–TlBr–TlI, Tl<sub>2</sub>Te–TlCl–TlI, Tl<sub>2</sub>Te–TlCl–TlBr systems were studied by means of DTA and X-ray analysis, as well as by measurement of microhardness and EMF of the concentration chains relative to the thallium electrode [141,157,158]. Some polythermal sections, an isothermal section at 300K, as well as the projection of the liquidus surface of all three systems were constructed. The primary crystallization areas, types and coordinates of non- and monovariant equilibria determined.

Tl<sub>5</sub>Te<sub>2</sub>Hal–Tl<sub>5</sub>Te<sub>2</sub>Hal' systems are nonquasi-binary due to the incongruent melting character of ternary compounds and consists of aliquidation fields. However, isostructural Tl<sub>5</sub>Te<sub>2</sub>Hal (Hal – Cl, Br, I) compounds form continuous solid solutions in solid state. Thus, a very interesting and rare phenomenon is observed in these systems – despite the fact that, there is a complete aliquidation in liquid state, the initial constituents of these systems are completely dissolved in each other in solid state. Realization of a such an unusual phenomenon in the Tl<sub>5</sub>Te<sub>2</sub>Hal–Tl<sub>5</sub>Te<sub>2</sub>Hal' systems is related to the fact that the aliquidation in these systems

does not reflect the character of interaction between the ternary compounds, but rather the melting character of these compounds and solid solutions based on them [159]. Also the composition of the layered liquid phases is outside of the T-x plane of these systems.

In all  $Tl_2Te-TlHal-TlHal'$  systems, the homogeneity areas of the solid solutions based

on the  $Tl_5Te_2Hal$  compounds fall outside the  $Tl_5Te_2Hal-Tl_5Te_2Hal'$  sections and occupy a wide area in the  $Tl_2Te-Tl_5Te_2Hal-Tl_5Te_2Hal'$  subsystems. This is due to the existence of a large one-sided homogeneity area on the basis of  $Tl_5Te_2Hal$  compounds in the  $Tl_2Te-TlHal$  systems.

## 6. Phase equilibria in some systems consisting of compounds of the $Tl_5Te_3$ type crystal structure

$Tl_5Te_3-Tl_9BiTe_6-Tl_5Te_2Hal$  systems are of great interest in terms of preparation of thermoelectric materials with very low heat conductivity. The point is that one of the components -  $Tl_9BiTe_6$  compound has a record high thermoelectric efficiency [29]. By incorporating halogen atoms in the crystal lattice of this compound, it is possible to further complicate them and reduce the thermal conductivity.

Phase equilibria in the  $Tl_5Te_3-Tl_9BiTe_6-Tl_5Te_2Hal$  systems were investigated by physico-chemical analysis methods, the isothermal sections at 760 and 800K and the projections of the liquidus surfaces of these systems were constructed [160-163].

$Tl_5Te_3-Tl_9BiTe_6$  and  $Tl_5Te_3-Tl_5Te_2Hal$  boundary systems of the  $Tl_5Te_3-Tl_9BiTe_6-Tl_5Te_2Hal$  concentration triangles were studied in [83,89,164]. It showed that these systems are characterized by unlimited dissolution of components one into another.  $Tl_9BiTe_6-Tl_5Te_2Hal$  systems form a continuous solid solutions ( $\delta$ -phase) and are quasi-binary due to the incongruent melting of tellurohalides

[163]. The composition of liquid phases go beyond the T-x diagram of these systems. Powder diffractograms of all alloys have a  $Tl_5Te_3$  type tetragonal structure while the dependence of lattice parameters on the composition is almost linear.

The liquidus surfaces of the  $Tl_5Te_3-Tl_9BiTe_6-Tl_5Te_2Hal$  systems are composed of two parts matching to the primary crystallization of  $\delta$ -phase from the homogeneous alloy and  $L+L_1$  immiscibility area. The  $ab$  curve connecting these two regions is characterized by the  $L+L_1 \rightarrow \delta$  monovariant syntectic equilibrium. The solidus of the system consists of a surface to comply with the end of crystallization (dashed line) [160-162]. The liquidus projection of all three systems is qualitatively similar and differs only by the size of the  $L+L_1$  immiscibility area. The results on the phase equilibrium can be used to select the composition of the sample and temperature regimes when growing  $\delta$ -solid solution crystals of given composition [159].

## Conclusion

In the review paper, the literature data and results of the work of the authors on physico-chemical research into some ternary and more complex systems which are considered important in terms of creation of physico-chemical bases for directional synthesis of thallium chalcogenides and related multicomponent non-stoichiometric phases. It revealed that as a result of the studies in these systems, a wide range of ternary compounds and multicomponent solid solutions of great

interest as promising functional materials with thermoelectric, optical, magnetic, sensor and detector properties were identified. These data open up great opportunities for optimizing the functional properties of phases by directional variation of their composition. Besides, the fundamental thermodynamic properties of thallium chalcogenides and many phases of variable composition which are valuable as scientific data and important for optimizing conditions of their design were determined.

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**TALLIUMUN XALKOHALOGENİDLƏRİ ƏSASINDA ÜÇLÜ VƏ DAHA MÜRƏKKƏB FAZALARIN  
ALINMASININ FİZİKİ-KİMYƏVİ ASPEKTLƏRİ**

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*İcmal məqalədə talliumun xalkohalogenidlərinin və onlar əsasında yeni çoxkomponentli fazaların alınması baxımından maraqlı kəsb edilən bəzi üçlü və daha mürəkkəb sistemlərin fiziki-kimyəvi tədqiqinin nəticələri sistemləşdirilmiş və təhlil edilmişdir. Qeyri-üzvi maddələrin bu sinfi termoelektrik, optik, maqnit, sensor və detektor xassələrinə malik olan perspektivli funksional materiallar kimi elmi və praktiki maraqlı kəsb edir. Burada həmçinin, tallium xalkohalogenidlərinin və onlar əsasında dəyişən tərkibli aralıq fazaların kristal quruluş xüsusiyyətləri, termodinamik və bəzi fiziki-kimyəvi xassələrinə aid ədəbiyyat məlumatları araşdırılmışdır.*

**Açar sözlər:** tallium xalkohalogenidləri, funksional materiallar, faza diaqramı, bərk məhlullar, termodinamik xassələr, kristal quruluş.

**ФИЗИКО-ХИМИЧЕСКИЕ АСПЕКТЫ РАЗРАБОТКИ ТРОЙНЫХ И СЛОЖНЫХ ФАЗ НА  
ОСНОВЕ ХАЛЬКОГАЛОГЕНИДОВ ТАЛЛИЯ**

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*В обзоре систематизированы и проанализированы результаты физико-химических исследований тройных и более сложных систем, представляющих интерес с точки зрения получения халькогалогенидов таллия и новых многокомпонентных фаз на их основе. Этот класс неорганических веществ представляет научный и практический интерес как перспективные функциональные материалы с термоэлектрическими, оптическими, магнитными, сенсорными и детекторными свойствами. Также проанализированы литературные данные о кристаллической структуре, термодинамических и некоторых физико-химических свойствах халькогалогенидов таллия и фаз на их основе.*

**Ключевые слова:** халькогалогениды таллия, функциональные материалы, фазовая диаграмма, твердые растворы, термодинамические свойства, кристаллическая структура.